Growth and optical damage properties of In:Zn:LiNbO₃ waveguide substrate

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Received 13 December 2005, revised 28 September 2006, accepted 6 October 2006 Published online 15 December 2006

Key words In:Zn:LiNbO₃ crystal, optical damage, waveguide substrate, m-line method. **PACS** 42.70.Mp, 42.65.Wi, 42.25.Fx, 74.62.Dh

In:Zn:LiNbO₃ crystals doped with different indium concentrations were grown by Czochralski technique. The optical damage threshold value and ultraviolet-visible absorption spectra of the In:Zn:LiNbO₃ crystals were measured. The In:Zn:LiNbO₃ crystals were made into optical waveguide substrates using hexanedioic acid as proton exchange agent. The optical damage resistant ability of those optical waveguide substrates was investigated by the m-line method. The optical damage threshold values of In(2mol.%):Zn(3mol.%):LiNbO₃ crystal and optical waveguide substrate are two orders of magnitude higher than those of pure LiNbO₃.

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1 Introduction

Lithium niobate (LiNbO₃) single crystal is of great interest for several decades due to its excellent nonlinear optical, piezoelectric and optic-electric properties [1]. It has various applications such as optical switch, frequency conversion, optical modulators and so on [2]. Proton exchange (PE) waveguide in LiNbO₃ substrate is widely used in integrated optics for its strong optical nonlinearity, large beam confinement and simply fabrication process [3,4]. But pure LiNbO₃ crystal shows a serious drawback of optical damage when it is irradiated with the high-power laser beam [5,6], which is limiting its usability. Various LiNbO₃ crystals doped with different optical damage resistant impurities have been prepared by the Czochralski technique. The optical damage problem of pure LiNbO₃ crystal can be greatly reduced by doping with MgO [7], ZnO [8], In₂O₃ [9], or Sc₂O₃ [10], etc. In the present work, we propose to dope In₂O₃ and ZnO into LiNbO₃ to grow In:Zn:LiNbO₃ crystals. The concentration of Zn²⁺ or In³⁺ in LiNbO₃ is respectively lower than its single-doped threshold concentration and the concentration of In³⁺ is changeable. In this paper, the shift mechanism of the absorption edges and enhance mechanism of the optical damage resistant ability on the In³⁺ concentration are discussed.

2 Experimental

Preparation of crystal samples In(1mol.%):Zn:LiNbO₃, In(1.5mol.%):Zn:LiNbO₃, In(2mol.%):Zn:LiNbO₃ crystals were grown by the Czochralski method. The intermediate frequency furnace was used as the heater. The starting materials were Li₂CO₃, Nb₂O₅, In₂O₃, ZnO with the purity of 99.99%. The Li/Nb ratio of the melts to grow the congruent LiNbO₃ is 0.946. The concentration of Zn is 3mol.%. The concentrations of In³⁺ are

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1mol.%, 1.5mol.% and 2mol.%, respectively. The compositions of the samples were listed in table 1. To prepare polycrystalline materials, the raw materials were mixed thoroughly, calcinated at 850°C for 4h, and sintered at 1150°C for 4h. The pulling rate was 2mm/h, the rotation rate was 20rpm and the temperature gradient was 30-40K/cm. The diameters and heights of these crystals were about 30mm and 50mm, respectively. The as-grown crystals were polarized in another furnace where the temperature was hold at about 1200°C for 4h with a direct current of 5 mA/cm². Then the crystals were cut into wafers of $15 \times 10 \times 2$ mm³. Then all the samples were polished to optical grade. Using hexanedioic acid as the proton exchange resource, PE In:Zn:LiNbO₃ waveguide substrates were prepared.

Table 1The compositions of the samples.

Crystal	1#	$2^{\#}$	3#	LiNbO ₃	PE 1 [#]	PE 2 [#]	PE 3 [#]	PE LiNbO ₃
In ³⁺ (mol.%)	1.0	1.5	2.0	0	1.0	1.5	2.0	0
Zn ²⁺ (mol.%)	3.0	3.0	3.0	0	3.0	3.0	3.0	0



Fig. 1 Setup for testing photo damage resistant ability of $In:Zn:LiNbO_3$ crystals. LS: light shed; S: beam splitter; CL: lens; D: power detector; OS: observation screen.

He-Ne Pout H2 Waveguide K-Y

Fig. 2 Experimental arrangement of m-line method. A: light attenuator; B: a piece of glass; C: crack; L: Fourier lens; H_1 and H_2 : photoelectric multipliers; P_{in} and P_{out} : input prism and output prism; X-Y: X-Y recorder with double pens.

Ultraviolet-visible absorption spectra and optical damage measurements The ultraviolet-visible absorption spectra of In:Zn:LiNbO₃ crystals were measured with a CARYIE style ultraviolet- visible spectrometer. The range was from 300nm to 1000nm. The facula-distortion observing method was applied to measure the optical damage resistant ability of the In:Zn:LiNbO₃ crystals. The experimental setup was shown in figure 1. The Ar⁺ laser (λ =488.0nm) was employed as the light source, and the output intensity of the beam could be regulated. The beam transited the light shed and convex lens and then focused on the sample. The focula could be observed on the screen. The polarization direction of the beam was along the c-axis of the crystal. When the power intensity was low, there was no optical damage and the facula was round. When the power intensity achieved a certain value, the facula began to elongate along the c-axis of the crystal. The optical damage occurred. When the facula shape distorted, the power density of the output beam could be recorded by the power detector. This value is defined as the optical damage threshold (*R*), which is an important parameter to evaluate the optical damage resistant ability of the crystal.

The optical damage resistant ability of the In:Zn:LiNbO₃ waveguide substrates were measured by the mline technique. The optical path was shown in figure 2. The He-Ne laser operated at 632.8nm was adopted as the light source. The waveguide and the input part of the rutile prism system were placed at the focus of the Fourier lens (f=250mm). The light transit through the waveguide and then went through the output part. The central facula of the m-line was accepted by the photoelectric multiplier through the crack C. The intensity of the input and the output could be obtained by the X-Y recorder with double pens. The intensity of the incident light could be regulated by the light attenuator. At the same time, we could observe the change of the center facula of the m-line.

It could be seen from the experiments that the brightness of the center facula of the m-line increased followed with the increasing of the incident intensity. The output value increased simultaneously. When the incident intensity achieved a certain value, the output value decreased suddenly and the center facula elongated. Finally, the center facula changed to a bright line. The optical damage occurred. The process of the

optical damage was shown in figure 3. The optical damage threshold value of the waveguide substrate (E_0) was defined as the incident laser intensity irradiating on the waveguide when the center facula suddenly elongated.



Fig. 3 M-line pictures of PE In:Zn:LiNbO₃ waveguide substrate during optical damage.



Fig. 4 Ultraviolet-visible absorption spectra of In:Zn:LiNbO₃ crystals.

3 Results and discussion

Figure 4 shows the ultraviolet-visible absorption spectra of the In:Zn:LiNbO₃ crystals. It can be seen that the absorption edges of the In:Zn:LiNbO₃ crystal shift to ultraviolet compared with Zn:LiNbO₃ and the shift extent increases with the increasing of the In³⁺ concentration.

The principal structure of LiNbO₃ crystal is formed by oxygen octahedron. The optical absorption edge is decided by the valence-electron energy from the 2p-orbits of O^{2^-} to the 4d-orbits of Nb⁵⁺. So the valence-electronic state of O^{2^-} directly affects the site of the absorption edge. If the doping ions make the O^{2^-} polarization ability increase, the energy for the electron transition decreases and the absorption edge shifts to infrared. On the contrary, the absorption edge shifts to ultraviolet. The polarization ability can be expressed by Z^{*2}/r , where Z^* is the effective nuclear charge number and r is the radius of the ion. $Z^*=Z-\sum s$, where Z is the nuclear charge number and $\sum s$ is the shield factor. The polarization abilities of Nb⁵⁺, In³⁺ and Li⁺ are 44.8, 22.8 and 2.5, respectively. From the viewpoint of Li-vacancy model, there are Li⁺ vacancies in congruent pure LiNbO₃ crystal and the Li⁺ vacancies are filled by Nb⁵⁺ to form antisite defects (Nb_{Li})⁴⁺ [11,12]. In the case of In:Zn:LiNbO₃ crystals, when In³⁺ concentration is low, In³⁺ ions occupy (Nb_{Li})⁴⁺. When In³⁺ concentration achieves a certain value, the additional In³⁺ ions begin to occupy Nb-site. This leads to the polarization ability of O²⁻ decreasing and the electron transition energy increasing. So the absorption edges of In:Zn:LiNbO₃ crystal shift to ultraviolet with the increasing of the In³⁺ concentration.

The results of the optical damage resistant ability of In:Zn:LiNbO₃ crystals and substrates are shown in table 2 and table 3, respectively. As can be seen, no matter among the crystals or among the waveguide substrates, the optical damage resistant ability of In(2mol.%):Zn:LiNbO₃ is about two orders of magnitude higher than that of pure LiNbO₃. Principal role for the parameters of the photo-damage plays electron-phonon anharmonicity, which is determined by the antisite Nb (Nb_{Li})⁴⁺ and oxygen vacancies of LiNbO₃ crystal [13,14]. From the viewpoint of Li-vacancy model, when In³⁺ and Zn²⁺ concentration is low, In³⁺ and Zn²⁺ ions

occupy antisite Nb $(Nb_{Li})^{4+}$ and exist in the forms of $(In_{Li})^{2+}$ and $(Zn_{Li})^{+}$. When their concentration is above a certain value, In^{3+} and Zn^{2+} ions begin to occupy Nb-site and exit in the forms of $(In_{Nb})^{2-}$ and $(Zn_{Nb})^{3-}$. Their ability to capture electrons decrease and the photoconductivity of In(2mol.%):Zn:LiNbO₃ crystal increase rapidly. According to the Scalar expression [15], the photorefractive ability of the LiNbO₃ crystal increases with the decrease of the photoconductivity. So with the increasing of the photoconductivity, the optical damage resistant ability of In(2mol.%):Zn:LiNbO₃ increase obviously.

 Table 2
 Optical damage resistant ability (R) of In:Zn:LiNbO3 crystals.

Sample	1#	2#	3#	LiNbO ₃
$R (W/cm^2)$	$1.3 \times 10^{3} \pm 5$	$2.8 \times 10^{3} \pm 5$	$1.8 \times 10^4 \pm 10$	$1.9 \times 10^{2} \pm 1$

Table 3 Optical damage threshold value (E_0) of In:Zn:LiNbO₃ waveguide substrates.

Sample	PE 1 [#]	PE 2 [#]	PE 3 [#]	PE LiNbO ₃
$E_0 (\mathrm{J/cm}^2)$	8.9×10^4	1.5×10^{5}	5.7×10^{6}	3.1×10 ⁴

4 Conclusion

To solve the low optical damage resistant ability of LiNbO₃ crystal, In:Zn:LiNbO₃ crystals have been grown by doping In_2O_3 and ZnO in LiNbO₃ crystal. The In:Zn:LiNbO₃ waveguide substrates have been prepared by proton exchange. In the In:Zn:LiNbO₃ crystals and waveguide substrates, the optical damage resistant abilities of In(2mol.%):Zn:LiNbO₃ increase two orders of magnitude higher than those of pure LiNbO₃.

Acknowledgements The work was supported by the National Natural Science Foundation of China (50232030, 10572155, 10172030) and the Science Foundation of Guangzhou Province (2005A10602002).

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